

Mécanique des composites

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Mécanique des composites

Micro et macromécanique

Tests mécaniques

Endommagement et rupture

CADFEM

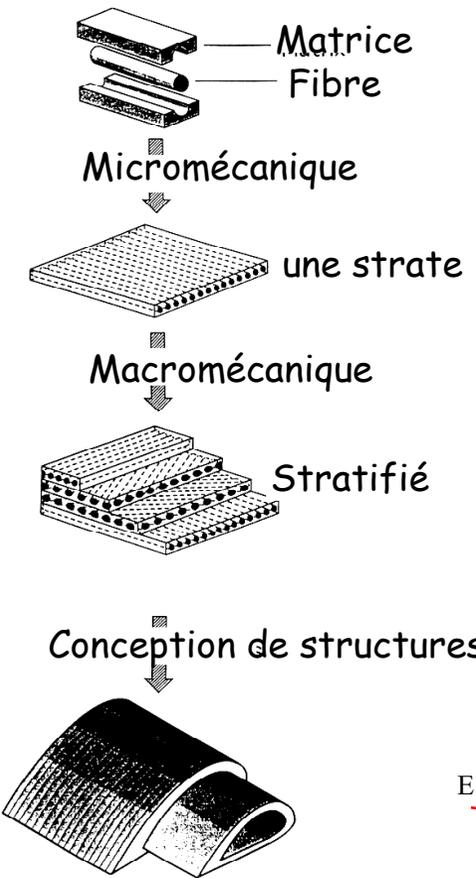
Applications

Les stratifiés

Structures sandwich

Composites textiles

De la fibre à la structure



$$E_1 = E_f V_f + E_m (1 - V_f) \quad P = \frac{P_m (1 + \xi \chi V_f)}{1 - \chi V_f}$$

$$Q_{11} = \frac{E_1}{(1 - \nu_{12} \nu_{21})} \quad \bar{Q}_{11} = m^4 Q_{11} + 2m^2 n^2 (Q_{12} + 2Q_{66}) + n^4 Q_{22}$$

$$\begin{bmatrix} \sigma_1 \\ \sigma_2 \\ \tau_{12} \end{bmatrix} = \begin{bmatrix} Q_{11} & Q_{12} & 0 \\ Q_{12} & Q_{22} & 0 \\ 0 & 0 & Q_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_1 \\ \varepsilon_2 \\ \gamma_{12} \end{bmatrix} \quad \begin{bmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{bmatrix} = \begin{bmatrix} \bar{Q}_{11} & \bar{Q}_{12} & \bar{Q}_{16} \\ \bar{Q}_{12} & \bar{Q}_{22} & \bar{Q}_{26} \\ \bar{Q}_{16} & \bar{Q}_{26} & \bar{Q}_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_x \\ \varepsilon_y \\ \gamma_{xy} \end{bmatrix}$$

$$A_{ij} = \sum_{k=1}^N (\bar{Q}_{ij})_k (z_k - z_{k-1})$$

$$\begin{bmatrix} N_x \\ N_y \\ N_{xy} \end{bmatrix} = \begin{bmatrix} A_{11} & A_{12} & A_{16} \\ A_{12} & A_{22} & A_{26} \\ A_{16} & A_{26} & A_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_x^0 \\ \varepsilon_y^0 \\ \gamma_{xy}^0 \end{bmatrix} + \begin{bmatrix} B_{11} & B_{12} & B_{16} \\ B_{12} & B_{22} & B_{26} \\ B_{16} & B_{26} & B_{66} \end{bmatrix} \begin{bmatrix} K_x \\ K_y \\ K_{xy} \end{bmatrix}$$

$$\begin{bmatrix} \varepsilon_x^0 \\ \varepsilon_y^0 \\ \gamma_{xy}^0 \end{bmatrix} = \begin{bmatrix} a_{11} & a_{12} & 0 \\ a_{12} & a_{22} & 0 \\ 0 & 0 & a_{66} \end{bmatrix} \left\{ \begin{bmatrix} N_x \\ N_y \\ N_{xy} \end{bmatrix} + \begin{bmatrix} N_x^T \\ N_y^T \\ 0 \end{bmatrix} \right\}$$

$$E_x = \frac{A_{11} A_{22} - A_{12}^2}{h A_{22}}$$

$$\begin{bmatrix} \varepsilon_x \\ \varepsilon_y \\ \gamma_{xy} \end{bmatrix} = \begin{bmatrix} \frac{1}{E_x} & -\frac{\nu_{xy}}{E_x} & 0 \\ -\frac{\nu_{yx}}{E_y} & \frac{1}{E_y} & 0 \\ 0 & 0 & \frac{1}{G_{xy}} \end{bmatrix} \begin{bmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{bmatrix}$$

Mx élastiques
Porosité nulle/minimale
Interfaces fibre-matrice parfaites/optimales

Orientations des fibres

Orthotropie
Contraintes planes

Adhésion entre plis parfaite/optimale

Symétries d'empilement
Effets de couplage

Cas de charges
Propriétés effectives
Mécanique des matériaux

Exemples

Orientation des fibres

Et si j'augmente le nombre de plis
tout en gardant la même épaisseur ?

Pourquoi des plis à 45 ?

Les hybrides

...

Contraintes et Rupture

Mécanique des matériaux

Contraintes principales

Critères de rupture

Rupture du premier pli

Rupture du composite

Endommagement

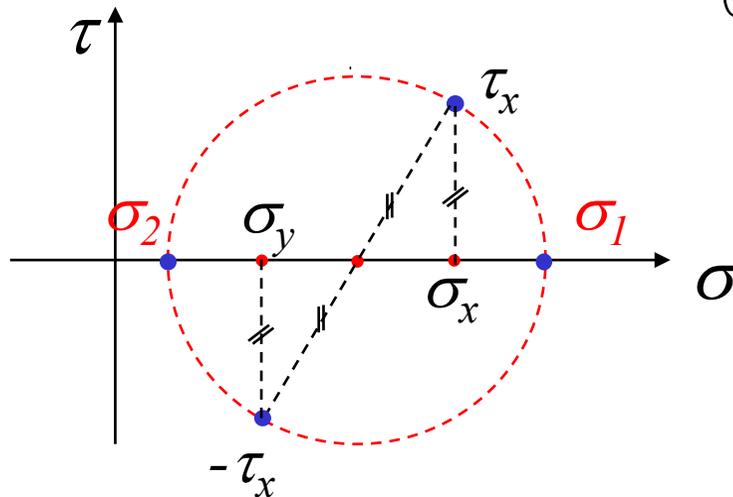
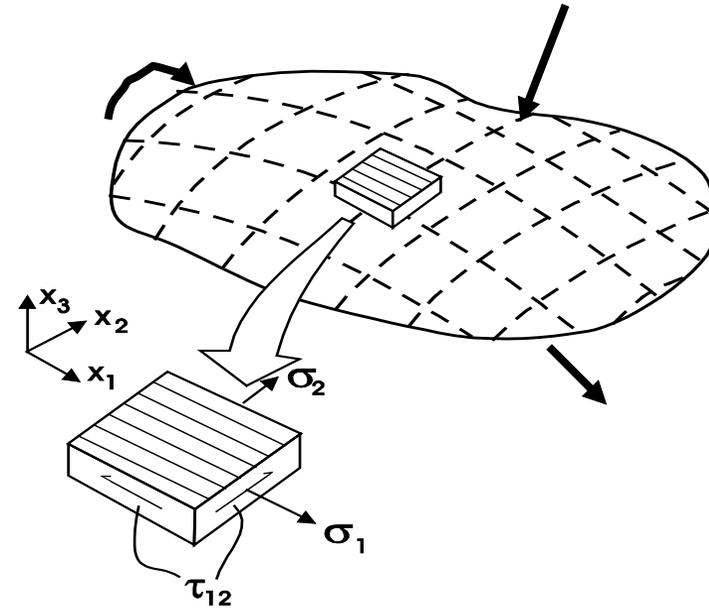
Efforts, contraintes et contraintes principales

$$\sigma = \frac{N}{S}$$

$$\sigma = \frac{M}{I_y} \cdot z$$

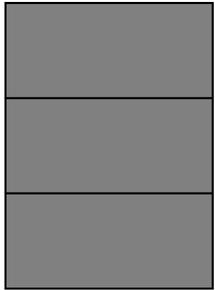
$$\frac{1}{r_c} = \frac{M}{E \cdot I}$$

$$\tau = \frac{M_t}{I_P} \cdot r$$

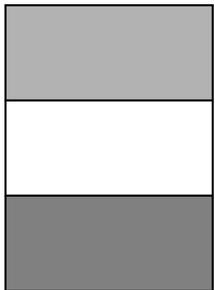
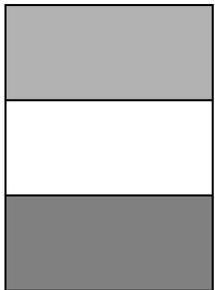


Idéalement il faudrait orienter les fibres selon les isostatiques.

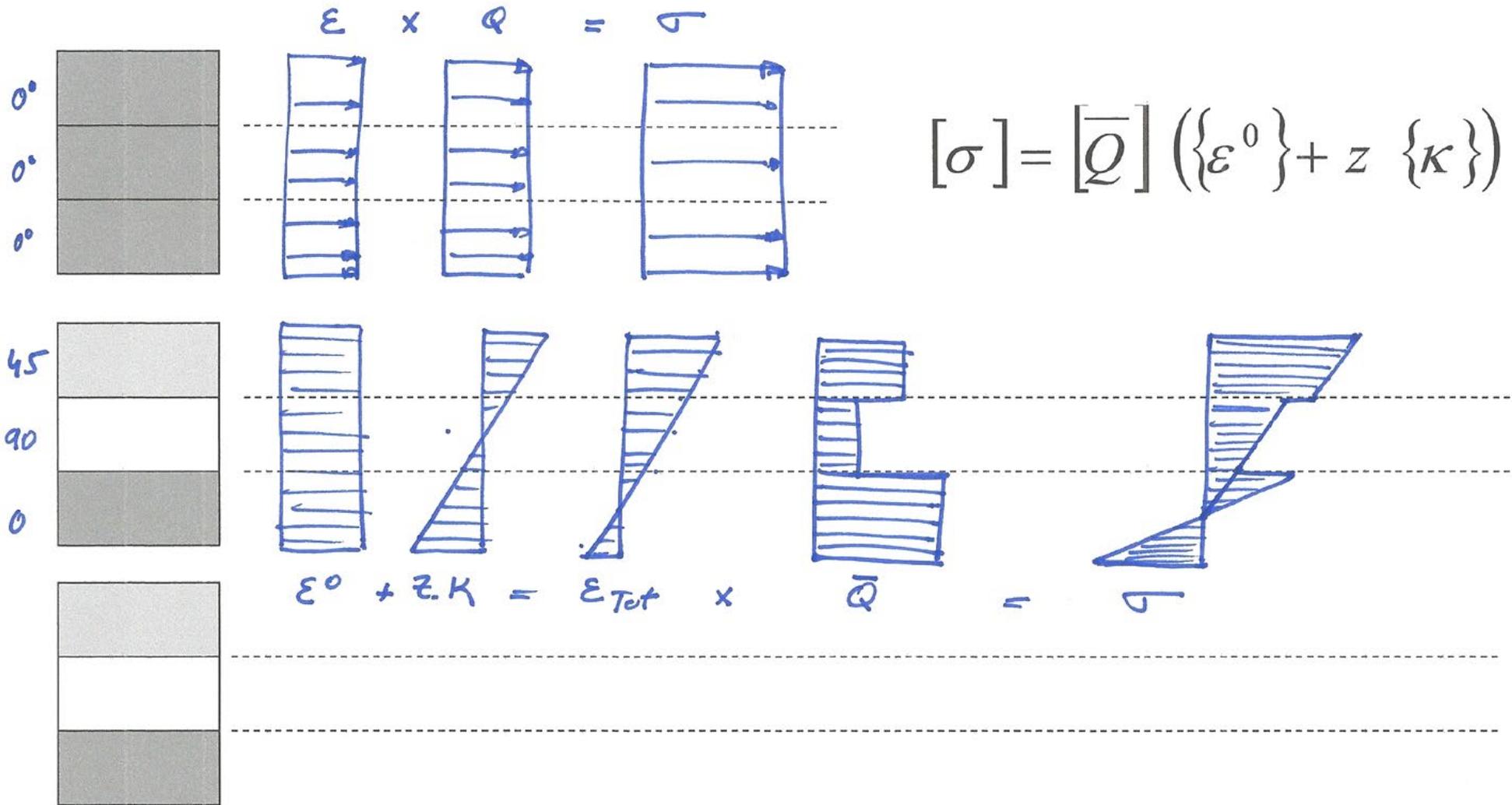
Contraintes dans les stratifiés anisotropes



$$[\sigma] = [\bar{Q}] \left(\{\varepsilon^0\} + z \{\kappa\} \right)$$



Contraintes dans les stratifiés anisotropes



Résistances ultimes

Laminate FPF analysis

Laminate : **C 04590S**

Modified : Sun Nov 11 17:28:27 2012

Lay-up : (0a/+45a/-45a/90a)SE h = 1.84 mm

| Ply | t | E_1 | E_2 | G_12 | nu_12 | G_31 | G_23 | X_t | X_c | Y_t | Y_c | S | R | Q | X_eps,t | X_eps,c | Y_eps,t | Y_eps,c | S_u |
|--------------------------|------|-----|-----|------|-------|------|---------|-----|-----|-----|-----|-----|-----|---------|---------|---------|----------|---------|------|
| | mm | GPa | GPa | GPa | | GPa | GPa | MPa | % | % | % | % | % |
| a E;Epoxy;UD-.230/299/50 | 0.23 | 38 | 9 | 3.6 | 0.3 | 3.6 | 3.46154 | 930 | 570 | 33 | 110 | 70 | 70 | 41.5385 | 2.44737 | 1.5 | 0.366667 | 1.22222 | 1.94 |

Load : **5kN10cm Mx125sur25cm**

Modified : Sun Nov 11 21:02:20 2012

Type : Forces and moments (Var.;E)

| | |
|-----------------|----------------|
| N_x = 50000 N/m | M_x = 500 Nm/m |
| N_y = 0 N/m | M_y = 0 Nm/m |
| N_xy = 0 N/m | M_xy = 0 Nm/m |
| Q_x = 0 N/m | |
| Q_y = 0 N/m | |

Factor of safety : FoS^v = 1

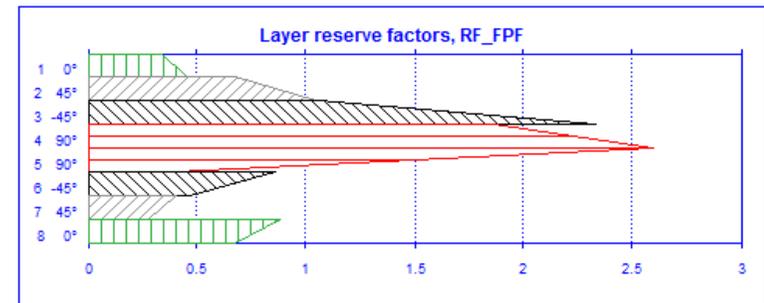
Failure criterion : Tsai-Wu; Max strain; Von Mises; Out-of-plane shear; Out-of-plane s (UD; non-UD; homogeneous; honeyc. core; foam/other core; adhe

Failure crit. param. : Tsai-Wu F_12*=-0.5

Stress/strain recovery : layer top/bottom

Laminate reserve factors

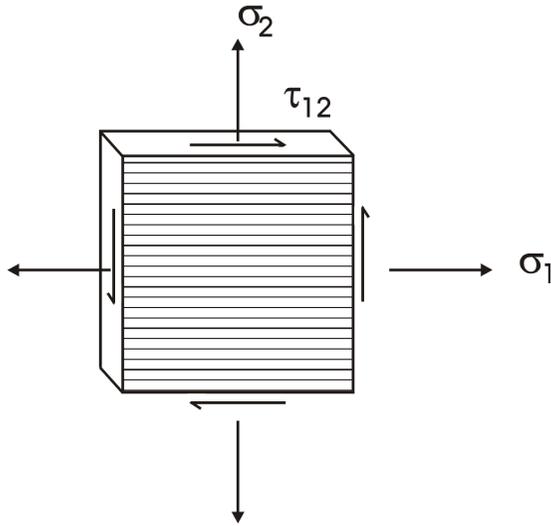
| FPF | Mode | FPF-only | Mode | Crit. layers | ILS | Crit. interf. |
|-----------|------|----------|------|--------------|-----|---------------|
| RF = 0.28 | 2t | 0.28 | 2t | 7(45°) | - | - |



Layer reserve factors - FPF

| Ply | theta | RF |
|-----|---------|---------------|
| 1 | a 0 t | 2. 0.34 1c/2t |
| | b | 0.45 |
| 2 | a 45 t | 6. 0.68 s |
| | b | 1.05 |
| 3 | a -45 t | 7. 1.06 s |
| | b | 2.33 |
| 4 | a 90 t | 8. 1.86 2c |
| | b | 2.60 |
| 5 | a 90 t | 3. 2.60 2t |
| | b | 0.40 |
| 6 | a -45 t | 4. 0.86 2t/s |
| | b | 0.47 |
| 7 | a 45 t | 1. 0.40 2t |
| | b | 0.28 |
| 8 | a 0 t | 5. 0.89 1t |
| | b | 0.67 |

Résistances ultimes



| | V_f | $\sigma_{u,1}^T$ | $\sigma_{u,2}^T$ | $\sigma_{u,1}^C$ | $\sigma_{u,2}^C$ | τ_u |
|--------------------------------------|-------|------------------|------------------|------------------|------------------|----------|
| carbone / époxy (AS/3501) | 0.60 | 1448 | 48 | 1172 | 248 | 62 |
| Kevlar 49 / époxy | 0.60 | 1380 | 27 | 276 | 65 | 60 |
| verre-E / époxy (scotch ply 1002) | 0.45 | 1100 | 28 | 620 | 138 | 83 |

Les normes

Table 2 Typical mechanical properties reported on data sheets

| Property | Test method | Test description | Usefulness for design purposes |
|-------------------------------|-------------|---|--|
| Tensile strength | ASTM D 638 | Property reports the stress in a standard specimen at the specified point on the stress-strain curve. Value reported could be yield or ultimate. | Depending on which value is reported, some portion of the tensile strength may be used as a design-stress limit. This is valid for gradual static loading at room temperature. For glass-reinforced materials, anisotropic properties render this property useless for design. |
| Elongation | ASTM D 638 | Property reports percent of elongation during the tensile test. | Property is meaningful for elongations from 0 to 15%. Data may be used to set interference limits or strain limits on a design. Strain rate, environmental temperature, and molding flaws such as knit lines must be considered in an actual part. For cases in which elongation exceeds 8%, the material typically has started to yield, and necking occurs. This is well beyond the point useful for design. |
| Tensile modulus | ASTM D 638 | Property reports the stiffness of the material. Value is the slope of the line tangent to the stress-strain curve at the origin. | Tensile modulus may be useful in calculating the deflection. Value is good only for small strains of 1% and at room temperature. Glass-reinforced materials behave anisotropically. Loads applied over extended time need to include creep. |
| Shear strength | ASTM D 732 | Property reports the shear stress calculated by dividing the load required to blank out the specimen by the area, as calculated by the perimeter of the punch times the specimen thickness. | The shear strength may have limited value in design engineering. Stress concentrations and shear rates that vary between the test and an application will affect the accuracy of the value. |
| Impact strength | ASTM D 256 | Property reports the energy absorbed when a notched specimen is subjected to a shock flexural loading. | Izod impact does not provide usable design information. It can indicate brittle behavior in a material, which would suggest certain approaches to design for impact loads. |
| Gardner impact | --- | Property reports the energy required for a projectile to fracture a sample plaque of the material. The projectile weight, shape, and support configuration are specified. | Gardner impact has very limited design value. The speed, shape, and weight of the projectile are likely to vary from actual use conditions. Also, the manner of support for a molded part differs from that used for the test plaque. |
| Tensile impact strength | ASTM D 1822 | Property reports the energy required to fracture a molded specimen subjected to a shock tensile load. | Tensile impact has little design value. It may indicate a brittle material that could affect the design approach employed. |
| Fracture toughness | --- | Property reports resistance to crack propagation. | Facilitates material selection |
| Flexural strength | ASTM D 790 | Property reports the maximum fiber stress in a simply supported bar loaded in flexure until either 5% strain is reached or the bar fractures. | Flexural strength has little design value. The value is based on classical elastic equations, which are inaccurate for large deflections, and are based on the assumption that the material obeys Hooke's Law in tension and compression. In actuality, the moduli are different. As with other strength properties, strain rate, temperature, and anisotropic properties limit the usefulness of this property in an application. |
| Flexural modulus | ASTM D 790 | Property reports the stiffness in flexure. Value is obtained by taking the slope of the tangent to the flexural stress-strain curve at origin. | Flexural modulus is useful as an approximation for the elastic modulus in deflection equations, or to convert strain into stress. However, it can be applied only to applications at room temperature and to small deflections. It is widely used in finite element analysis (FEA). Deflection is often the biggest problem in plastic part design. |
| Compression strength | ASTM D 695 | Property reports the maximum compressive stress carried by the test specimen during the compressive test. | Compressive strength cannot be considered significant for design when the application of the loads is widely different from the gradual test application. Fatigue and yielding are not considered and may need to be accounted for in an application. |
| Poisson's ratio | --- | --- | Poisson's ratio is needed for stress analysis, for both general deflection and FEA. |

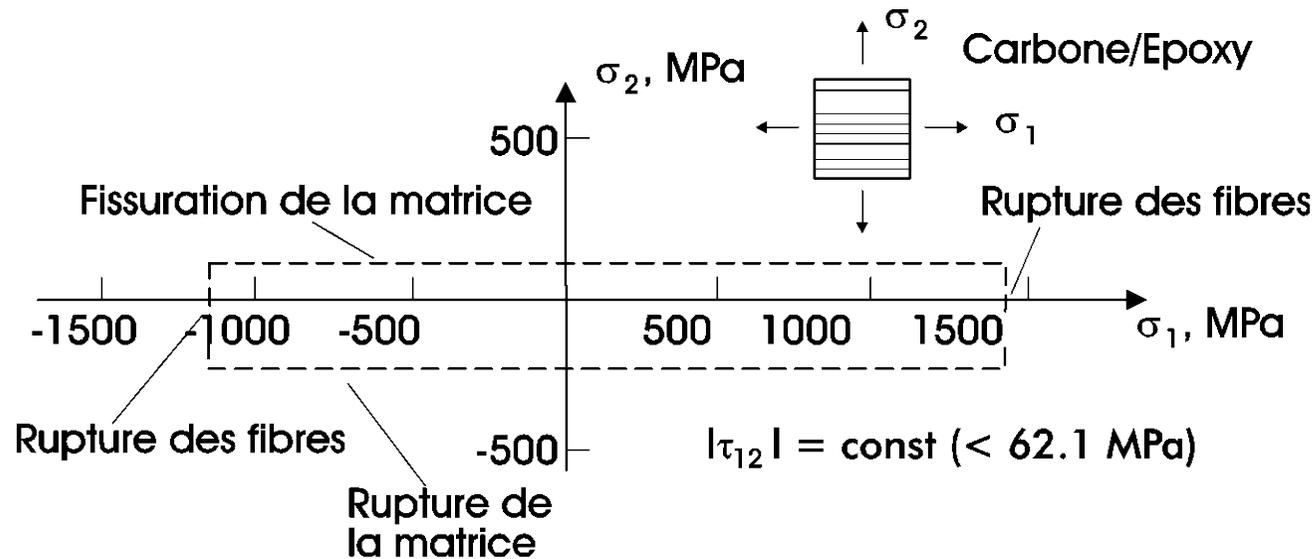
Critères de rupture

- Contraintes maximales
- Déformations maximales
- Critères d'interaction : Hill, Tsai-Hill
- Critère de Tsai-Wu
- Critères de Hashin

Critère de la contrainte maximale

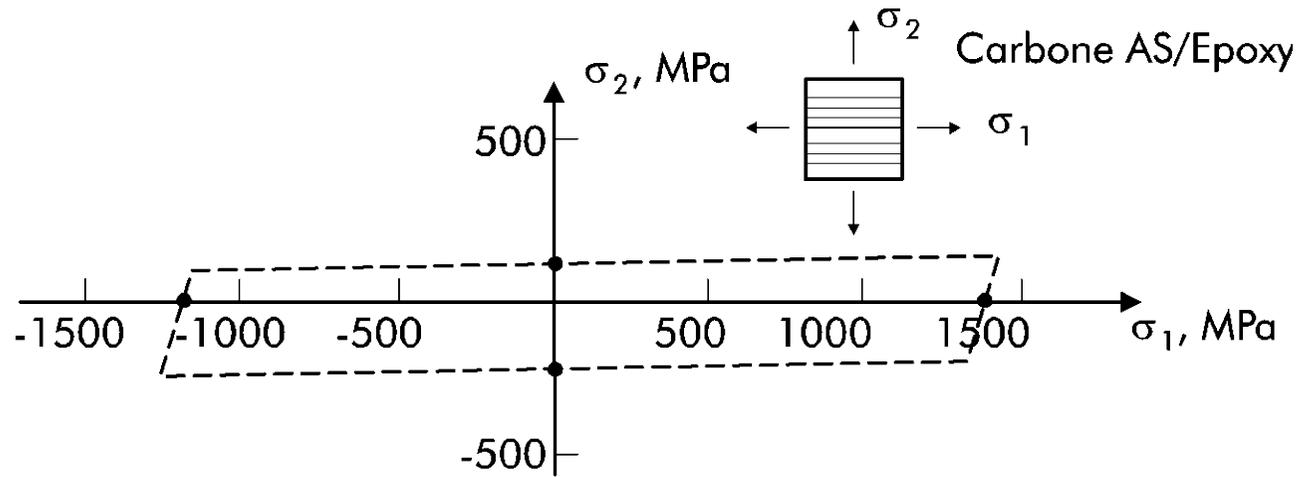
Traction et cisaillement $\sigma_1 = \sigma_{u,1}^T$ $\sigma_2 = \sigma_{u,2}^T$ $|\tau_{12}| = \tau_u$

Compression et cisaillement $|\sigma_1| = \sigma_{u,1}^C$ $|\sigma_2| = \sigma_{u,2}^C$ $|\tau_{12}| = \tau_u$



Critère de la déformation maximale

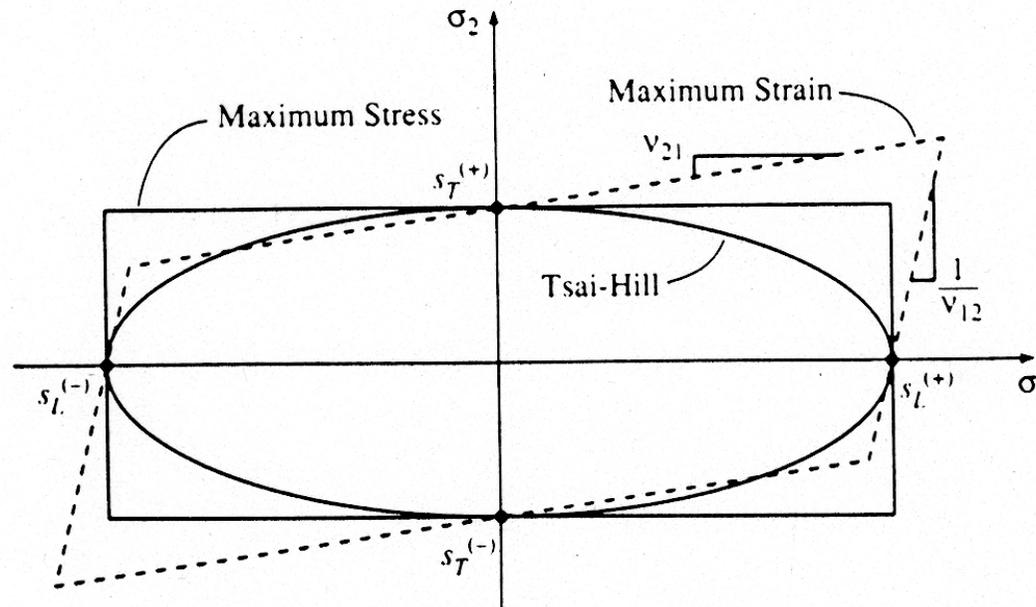
$$-\varepsilon_{u,i}^C < \varepsilon_i < \varepsilon_{u,i}^T$$



$$\sigma_{u,1} = \frac{\sigma_{u,2}^T - \sigma_2}{\nu_{21}} \quad \sigma_1 > 0$$

Critère de Tsai-Hill

$$\frac{\sigma_1^2 - \sigma_1\sigma_2}{(\sigma_{u,1}^T)^2} + \frac{\sigma_2^2}{(\sigma_{u,2}^T)^2} + \frac{\tau_{12}^2}{\tau_u^2} = 1$$



Critère de Tsai-Wu

$$F_1\sigma_1 + F_2\sigma_2 + F_{11}\sigma_1^2 + F_{22}\sigma_2^2 + F_{66}\tau_{12}^2 + 2F_{12}\sigma_1\sigma_2 = 1$$

$$F_1 = \frac{1}{\sigma_{u,1}^T} - \frac{1}{\sigma_{u,1}^C}$$

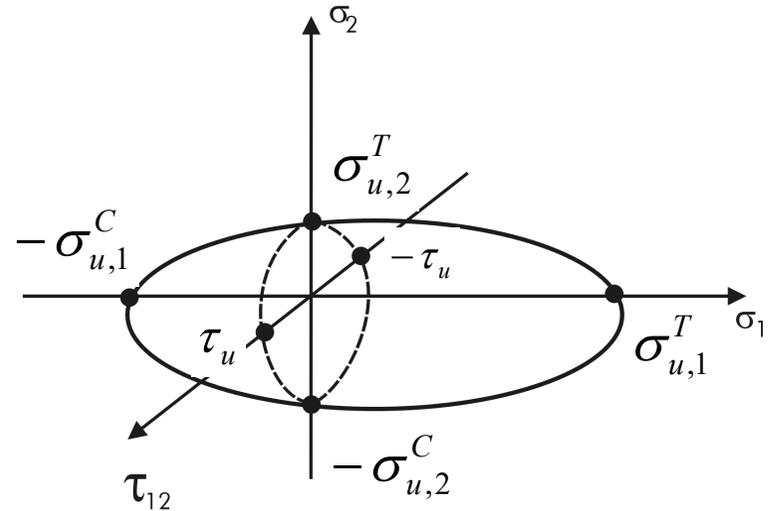
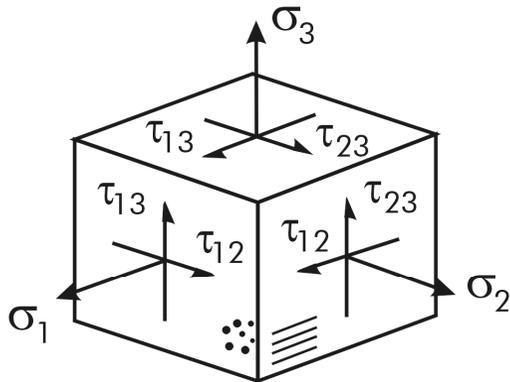
$$F_{11} = \frac{1}{\sigma_{u,1}^T \sigma_{u,1}^C}$$

$$F_{66} = \frac{1}{\tau_u^2}$$

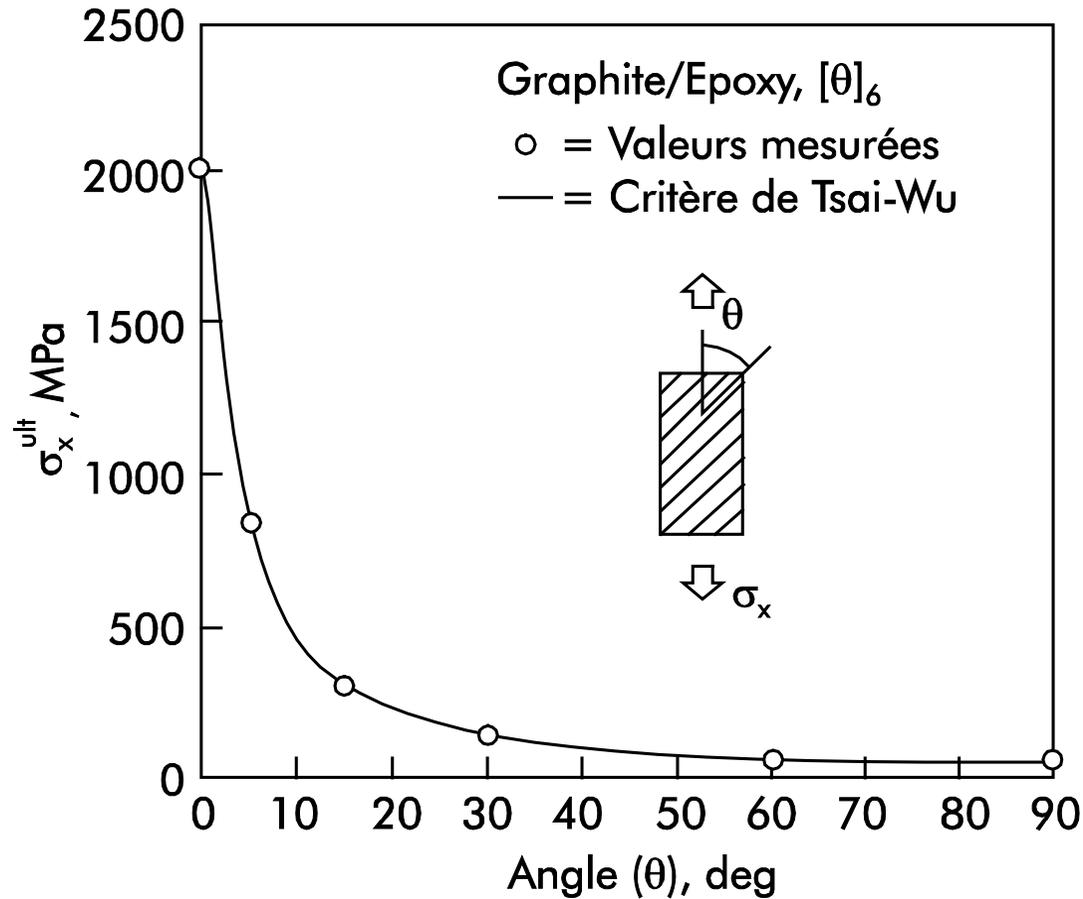
$$F_{12} = \frac{-\sqrt{F_{11}F_{22}}}{2}$$

$$F_2 = \frac{1}{\sigma_{u,2}^T} - \frac{1}{\sigma_{u,2}^C}$$

$$F_{22} = \frac{1}{\sigma_{u,2}^T \sigma_{u,2}^C}$$

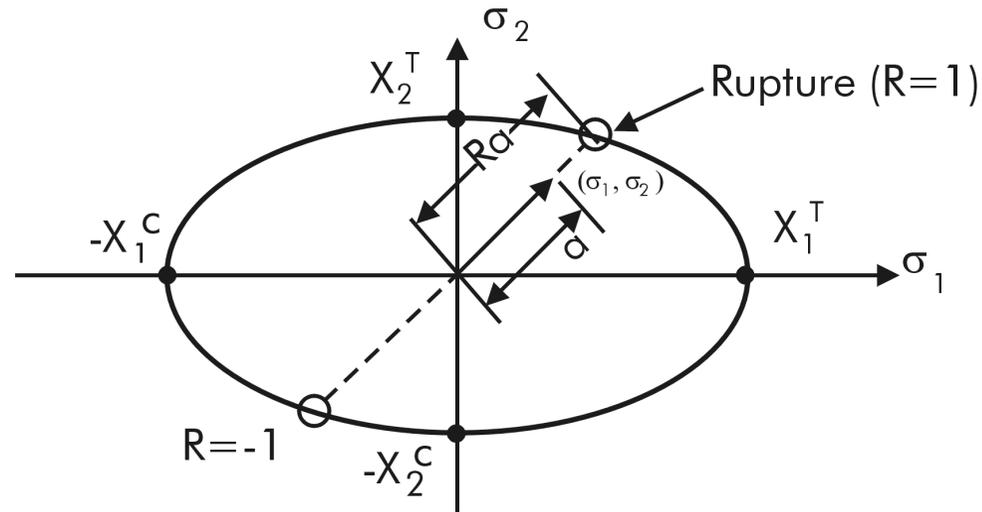


Critère de Tsai-Wu



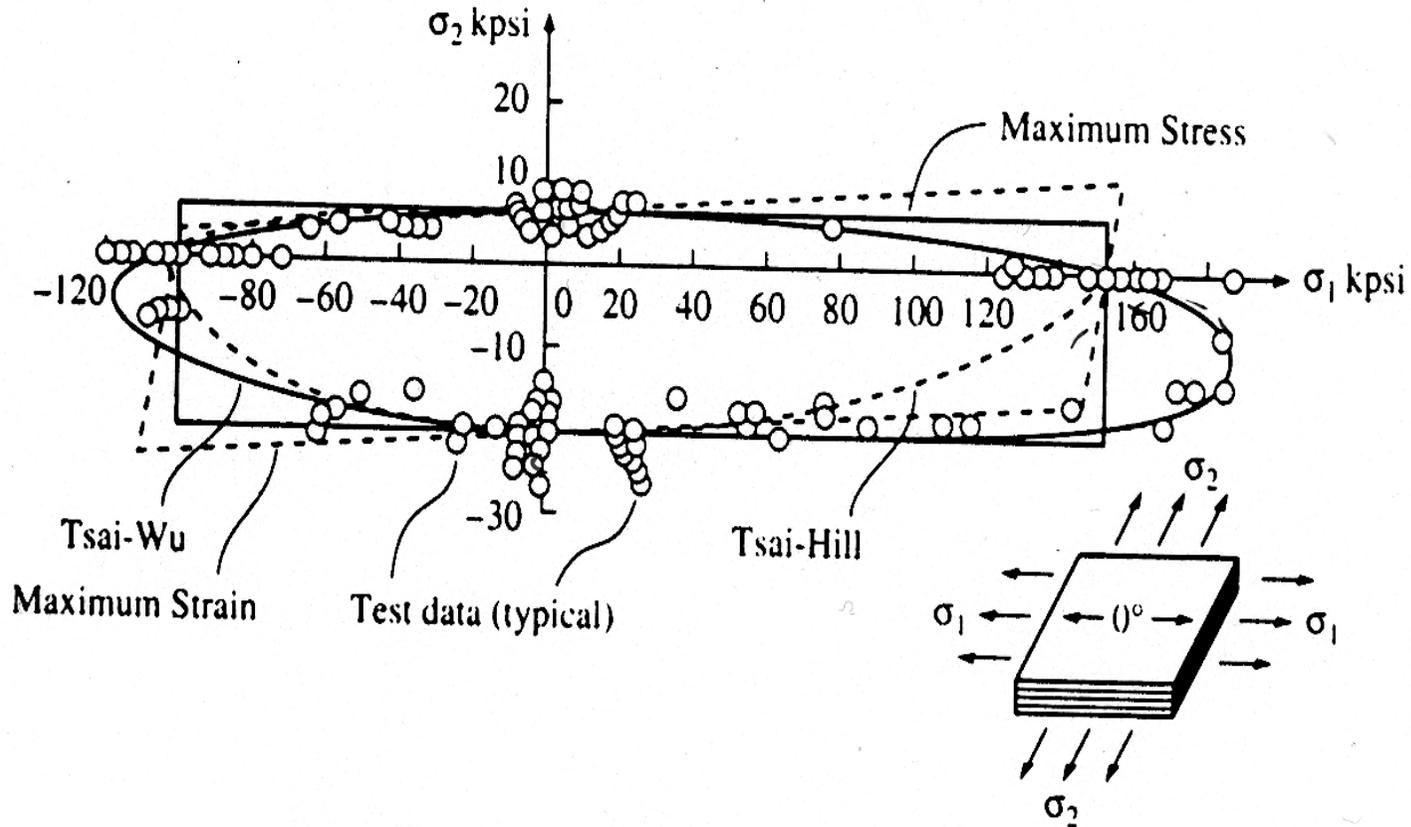
Facteur de sécurité

Reserve Factor



Mathématiquement, la valeur de R est calculée pour n'importe quel état de contraintes appliquées $(\sigma_1, \sigma_2, \tau_{12})$ par substitution de $(R\sigma_1, R\sigma_2, R\tau_{12})$ dans la théorie de la rupture adéquate et en résolvant pour R .

Critères de rupture



Rupture du premier pli puis rupture finale

Contraintes dans les strates

$$[\sigma] = [\bar{Q}] (\{\varepsilon^0\} + z \{\kappa\})$$

Nouvel état de contraintes

Critère de rupture

$$\frac{\sigma_1^2 - \sigma_1\sigma_2}{(\sigma_{u,1}^T)^2} + \frac{\sigma_2^2}{(\sigma_{u,2}^T)^2} + \frac{\tau_{12}^2}{\tau_u^2} = 1$$

Rupture d'un pli

$$(\bar{Q}_{ij})_{\text{pli rompu}} = 0$$

Re-calculation des constantes

$$A_{ij} = \sum_{k=1}^N (\bar{Q}_{ij})_k (z_k - z_{k-1})$$

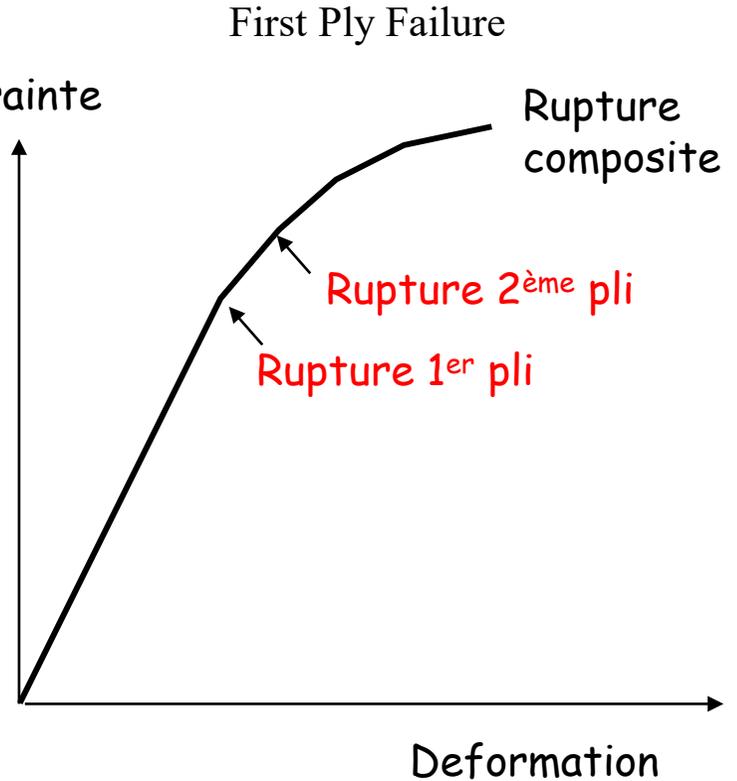
$$B_{ij} = \frac{1}{2} \sum_{k=1}^N (\bar{Q}_{ij})_k (z_k^2 - z_{k-1}^2)$$

$$D_{ij} = \frac{1}{3} \sum_{k=1}^N (\bar{Q}_{ij})_k (z_k^3 - z_{k-1}^3)$$

$$\begin{bmatrix} N_x \\ N_y \\ N_{xy} \end{bmatrix} = \begin{bmatrix} A_{11} & A_{12} & A_{16} \\ A_{12} & A_{22} & A_{26} \\ A_{16} & A_{26} & A_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_x^0 \\ \varepsilon_y^0 \\ \gamma_{xy}^0 \end{bmatrix} + \begin{bmatrix} B_{11} & B_{12} & B_{16} \\ B_{12} & B_{22} & B_{26} \\ B_{16} & B_{26} & B_{66} \end{bmatrix} \begin{bmatrix} \kappa_x \\ \kappa_y \\ \kappa_{xy} \end{bmatrix}$$

$$\begin{bmatrix} M_x \\ M_y \\ M_{xy} \end{bmatrix} = \begin{bmatrix} B_{11} & B_{12} & B_{16} \\ B_{12} & B_{22} & B_{26} \\ B_{16} & B_{26} & B_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_x^0 \\ \varepsilon_y^0 \\ \gamma_{xy}^0 \end{bmatrix} + \begin{bmatrix} D_{11} & D_{12} & D_{16} \\ D_{12} & D_{22} & D_{26} \\ D_{16} & D_{26} & D_{66} \end{bmatrix} \begin{bmatrix} \kappa_x \\ \kappa_y \\ \kappa_{xy} \end{bmatrix}$$

Contrainte



Dimensionnement : propriétés effectives

Efforts connus

$$\begin{bmatrix} N_x \\ N_y \\ N_{xy} \end{bmatrix} = \begin{bmatrix} A_{11} & A_{12} & A_{16} \\ A_{12} & A_{22} & A_{26} \\ A_{16} & A_{26} & A_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_x^0 \\ \varepsilon_y^0 \\ \gamma_{xy}^0 \end{bmatrix}$$

h = épaisseur du stratifié

Contraintes globales moyennes (fictives)

$$\begin{bmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{bmatrix} = 1/h \begin{bmatrix} N_x \\ N_y \\ N_{xy} \end{bmatrix} = 1/h \begin{bmatrix} A_{11} & A_{12} & A_{16} \\ A_{12} & A_{22} & A_{26} \\ A_{16} & A_{26} & A_{66} \end{bmatrix} \begin{bmatrix} \varepsilon_x^0 \\ \varepsilon_y^0 \\ \gamma_{xy}^0 \end{bmatrix}$$

$$\frac{1}{h} A_{ij} = \sum_{k=1}^N (\overline{Q_{ij}})_k \frac{\text{épaisseur}_k}{h} = \sum_{k=1}^N (\overline{Q_{ij}})_k \text{pourcentage}_k$$

$$\begin{bmatrix} \varepsilon_x \\ \varepsilon_y \\ \gamma_{xy} \end{bmatrix} = h \cdot \begin{bmatrix} A_{11} & A_{12} & A_{16} \\ A_{12} & A_{22} & A_{26} \\ A_{16} & A_{26} & A_{66} \end{bmatrix}^{-1} \begin{bmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{bmatrix} = \begin{bmatrix} \frac{1}{E_x} & -\frac{\nu_{xy}}{E_x} & 0 \\ -\frac{\nu_{yx}}{E_y} & \frac{1}{E_y} & 0 \\ 0 & 0 & \frac{1}{G_{xy}} \end{bmatrix} \begin{bmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{bmatrix}$$

Modules effectifs

Dimensionnement : déformation et épaisseur minimale

Choisir des pourcentages initiaux

Calculer $\frac{1}{h} A_{ij} = \sum_{k=1}^N (\overline{Q_{ij}})_k \text{pourcentage}_k$

Inverser pour obtenir modules effectifs $E_x, E_y \dots$ et les déformations

Critère de rupture pour le pli k en entrant dans le critère

de rupture $h \sigma_x$ qui est connu car = N_x

$$\frac{(h\sigma_1)^2 - h\sigma_1 h\sigma_2}{(\sigma_{u,1}^T)^2} + \frac{(h\sigma_2)^2}{(\sigma_{u,2}^T)^2} + \frac{(h\tau_{12})^2}{\tau_u^2} = h_k^2$$

h pour éviter la rupture du pli k est donc connu

idem pour autres plis

Choisir le $h = \sup(h_k)$

Pourcentages initiaux technologiques

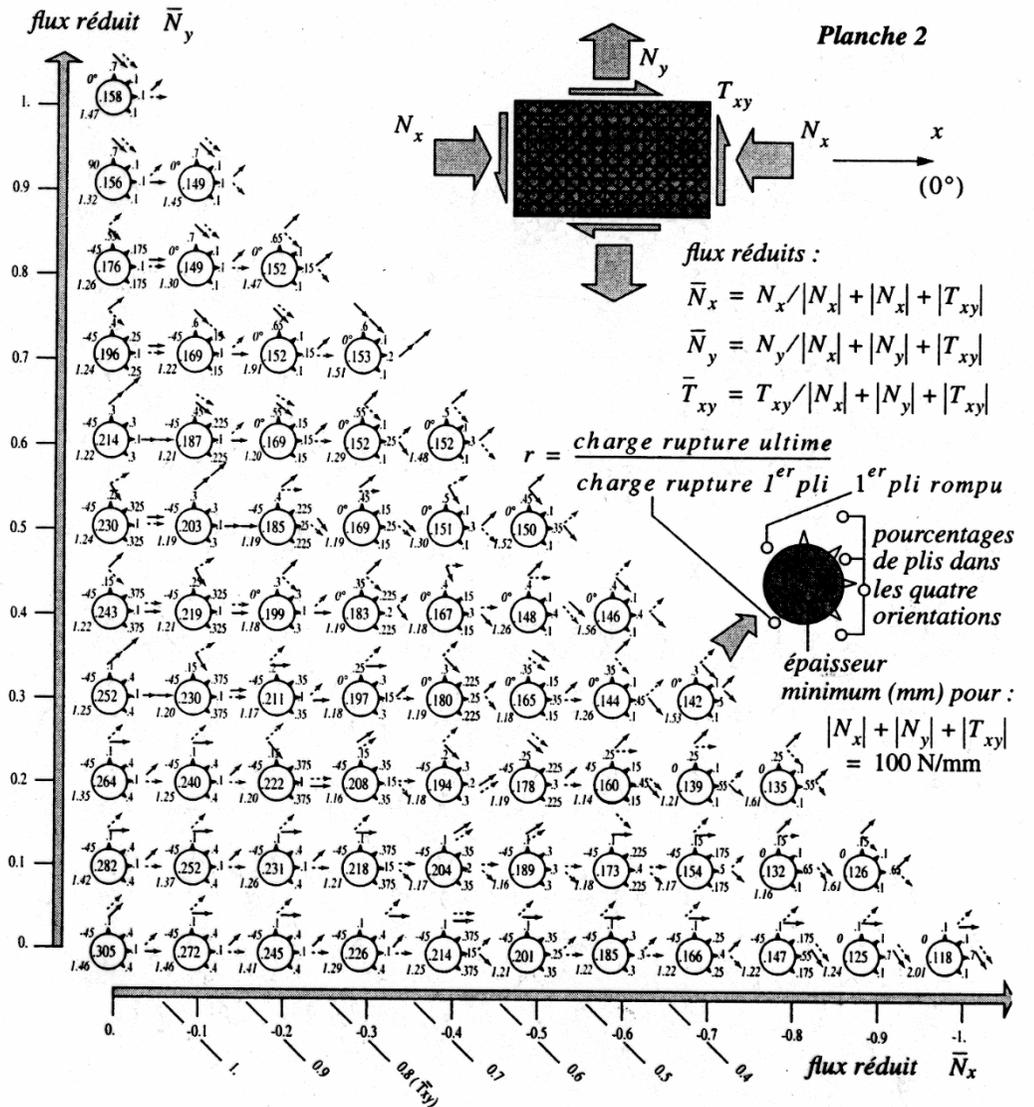
- Proportion minimum de 5 à 10% dans chaque direction 0, 90, +45, -45
- Epaisseur minimale du stratifié de l'ordre du mm (8 couches UD ou 3 à 4 couches de tissu équilibré)
- Considérer les directions des efforts principaux
- Plis à 90 placés en surface, puis plis à 45, puis à 0 lorsque les efforts prépondérants sont à 0
- Pas plus de 4 plis consécutif dans une même direction

Guides de conception

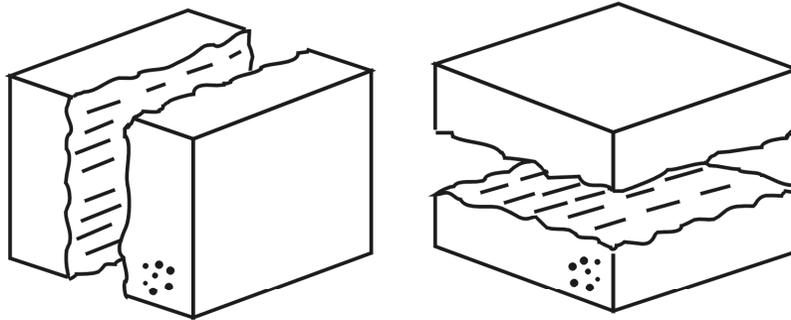
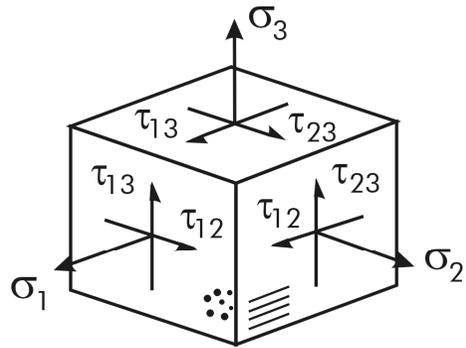
Composition optimum d'un stratifié carbone/époxyde

$V_f = 0,6$; caractéristiques du pli : cf. annexe 1 ou paragraphe 3.3.3.

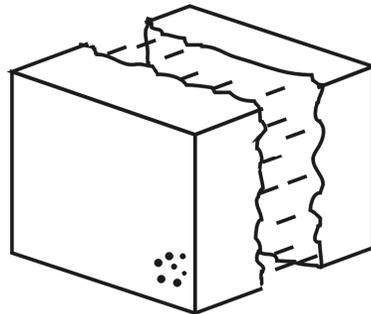
10 % minimum de plis dans chaque direction 0° , 90° , $+45^\circ$, -45°



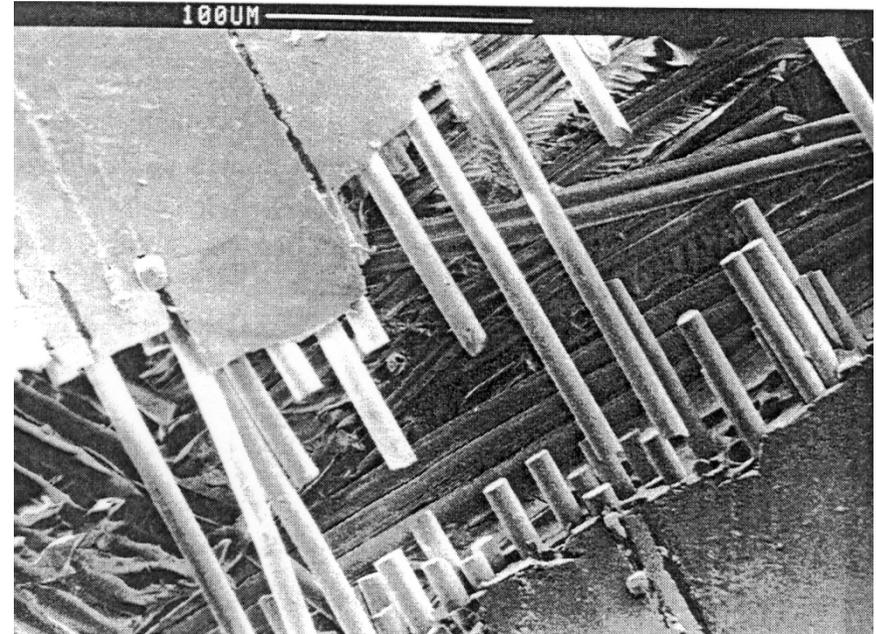
Endommagement



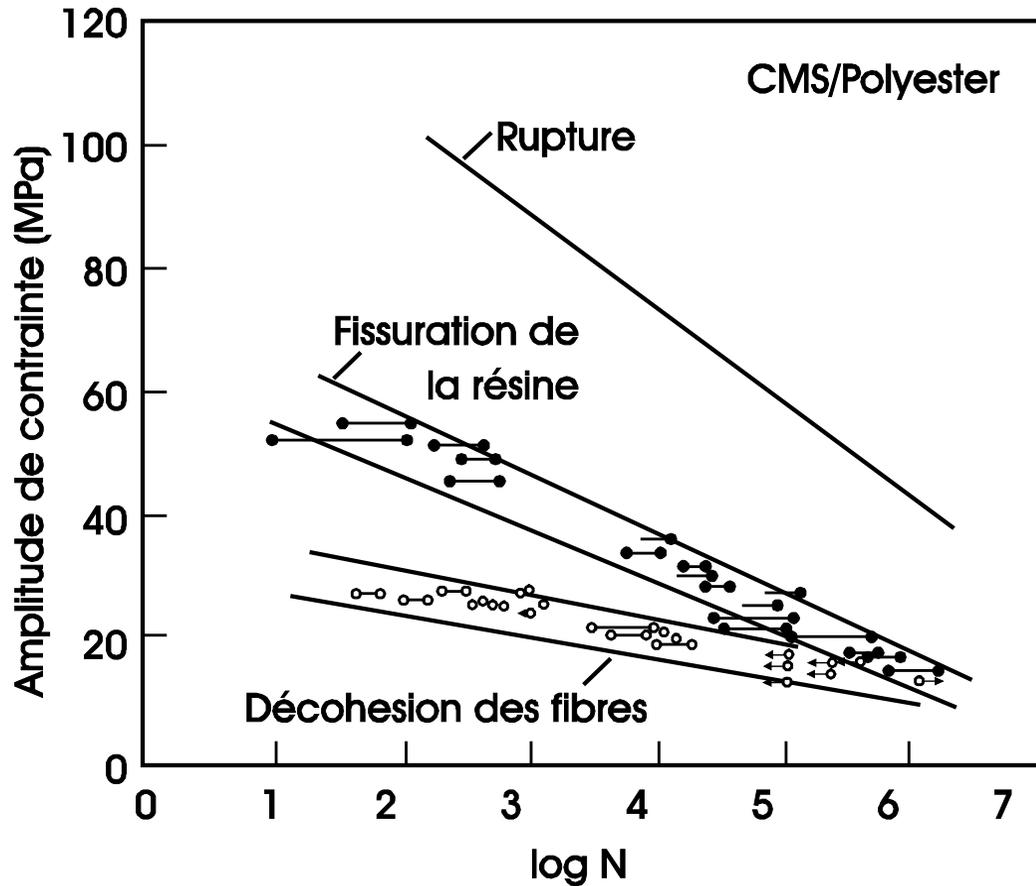
(b) Rupture de la matrice



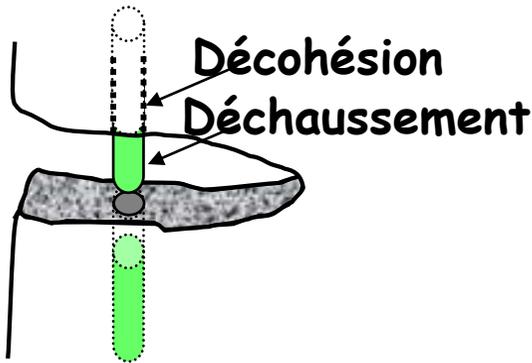
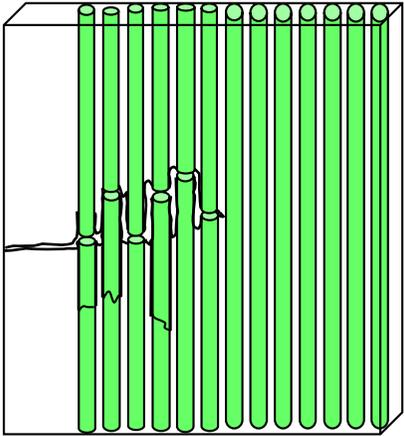
(c) Arrachage/rupture des fibres



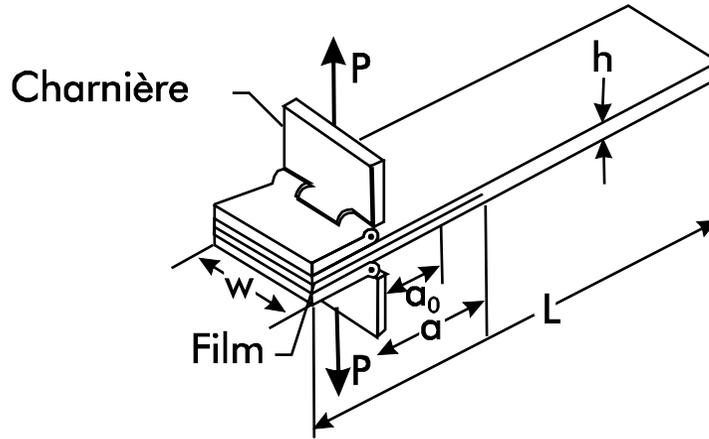
Fatigue des composites



Propagation de fissures



Energie de rupture typique



Approche G_{IC} , courbes R

d'une matrice thermodurcie: $\sim 100 \text{ J/m}^2$
 des fibres de verre: $\sim 10 \text{ J/m}^2$
 d'un composite: $\sim 10000 \text{ J/m}^2$

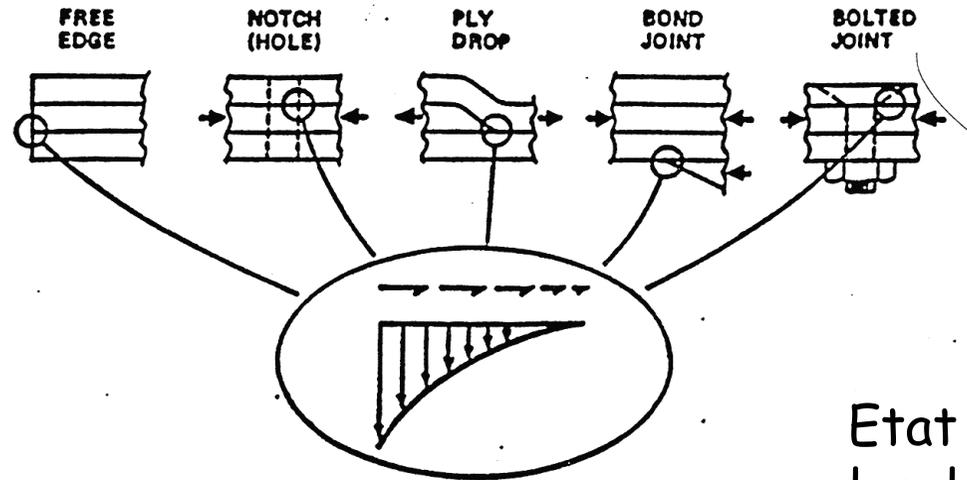
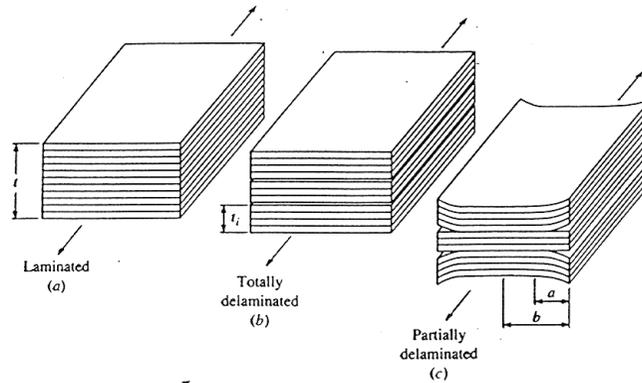
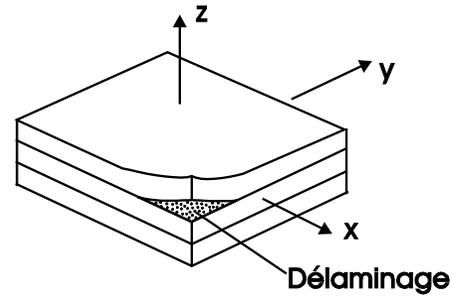


DLL-MAT-Mécanique-rupture - Quelle énergie propage une fissure ?
 Discovery Learning Videos

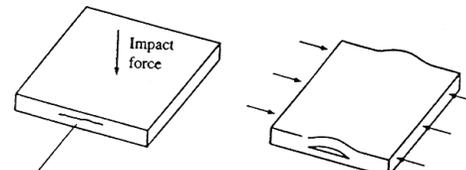
EPFL

<https://tube.switch.ch/videos/9333a14a>

Le délaminage



Etat de contraintes locales complexe



Mécanique des composites

Micro et macromécanique

Tests mécaniques

Endommagement et rupture

CADFEM

Applications

Les stratifiés

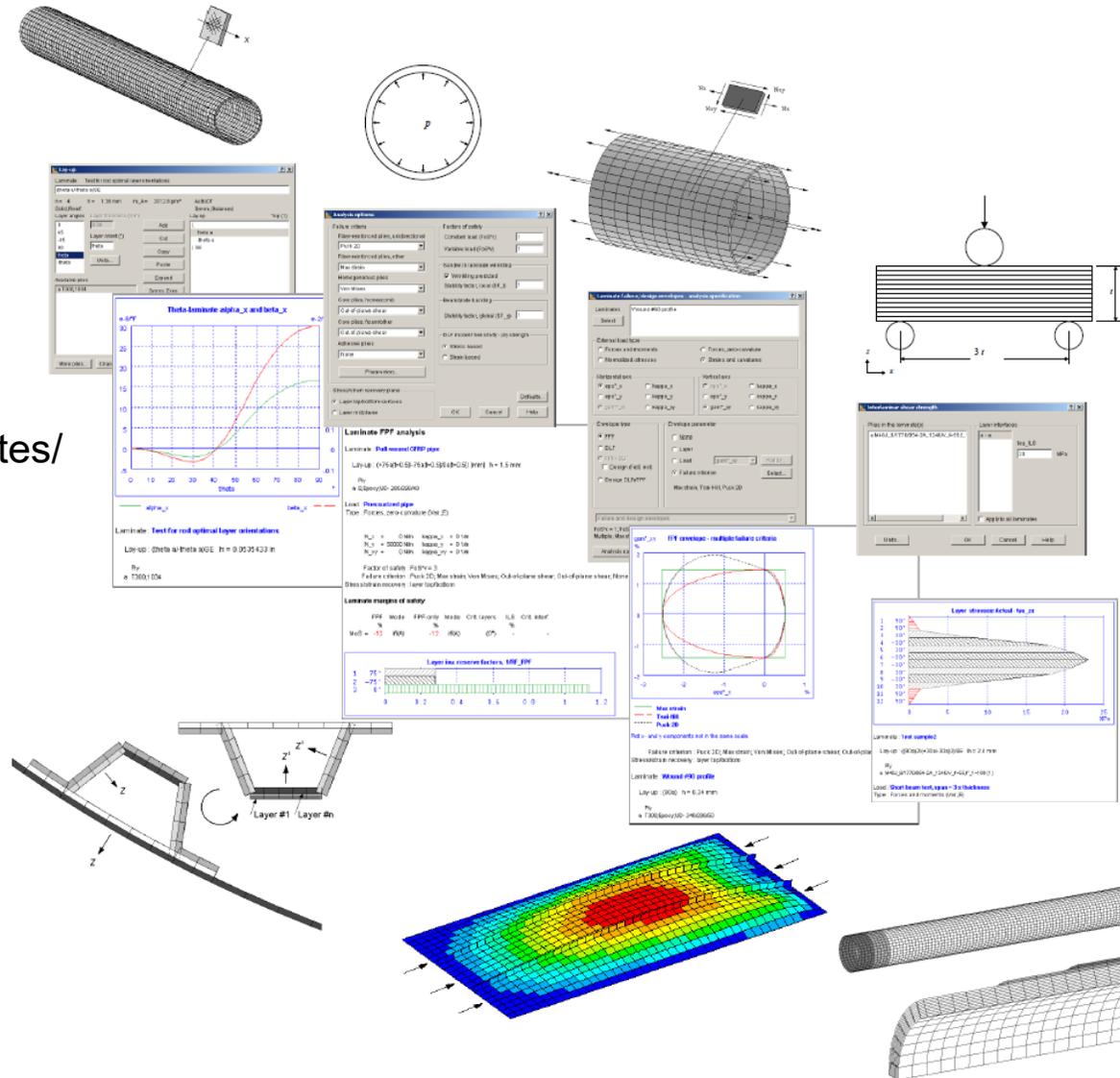
Structures sandwich

Composites textiles

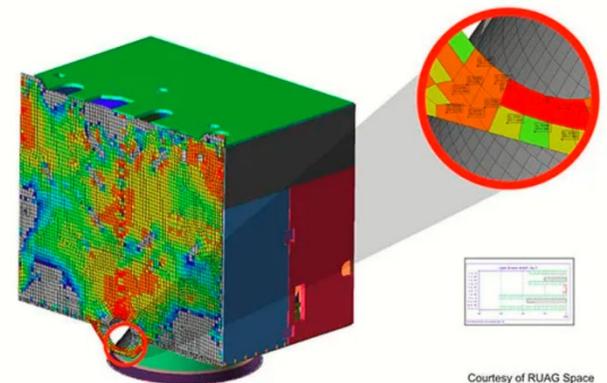
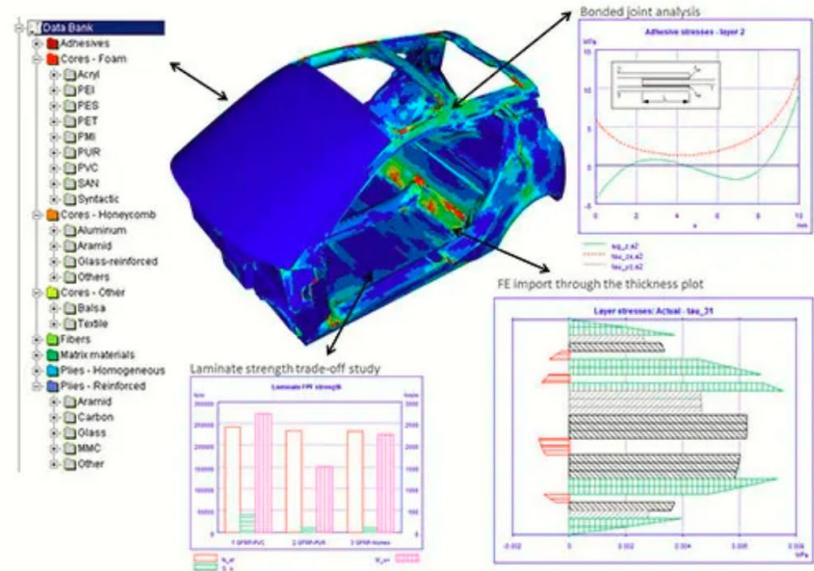
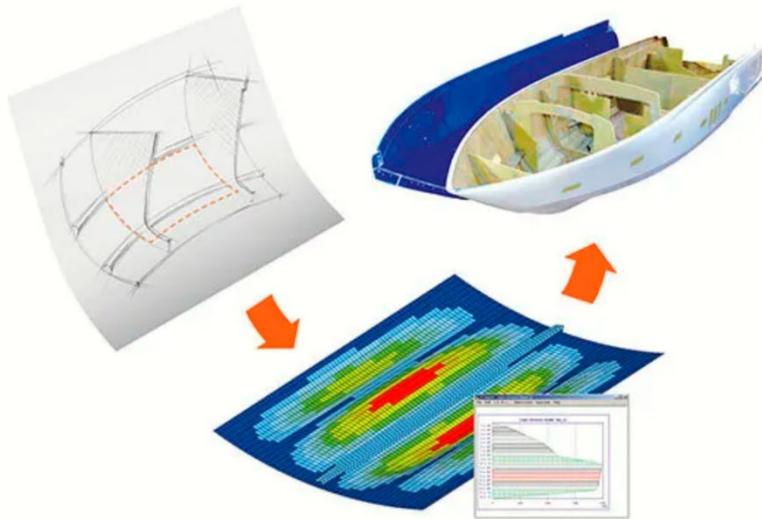
Mécanique des composites stratifiés



<https://www.altair.com/composites/>



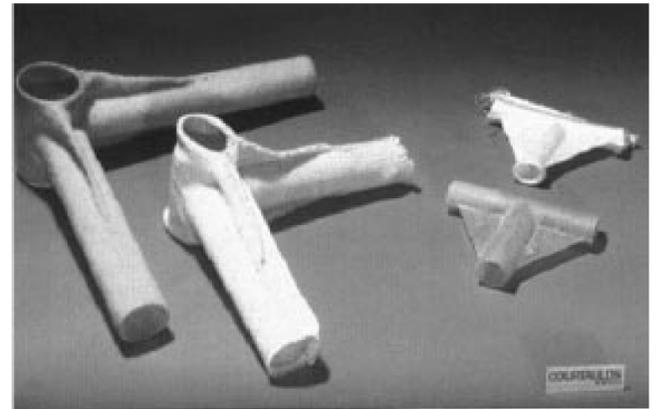
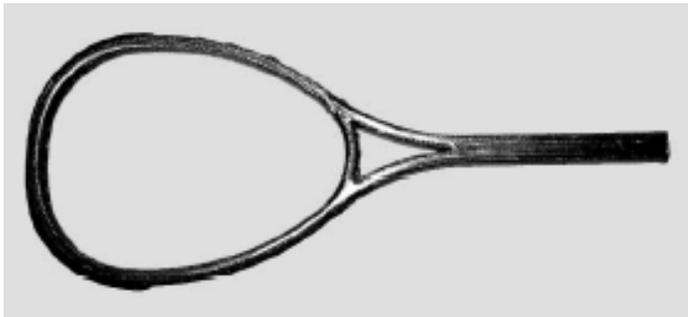
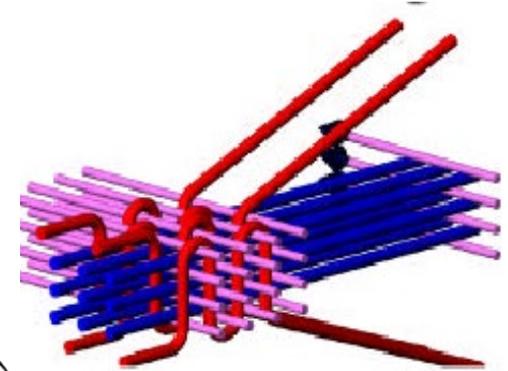
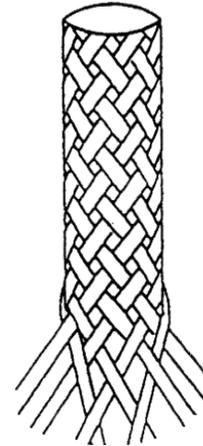
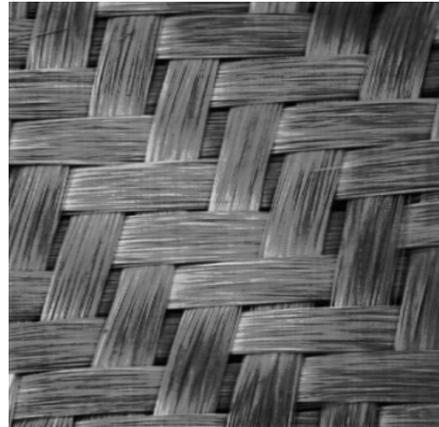
FEM



Courtesy of RUAG Space

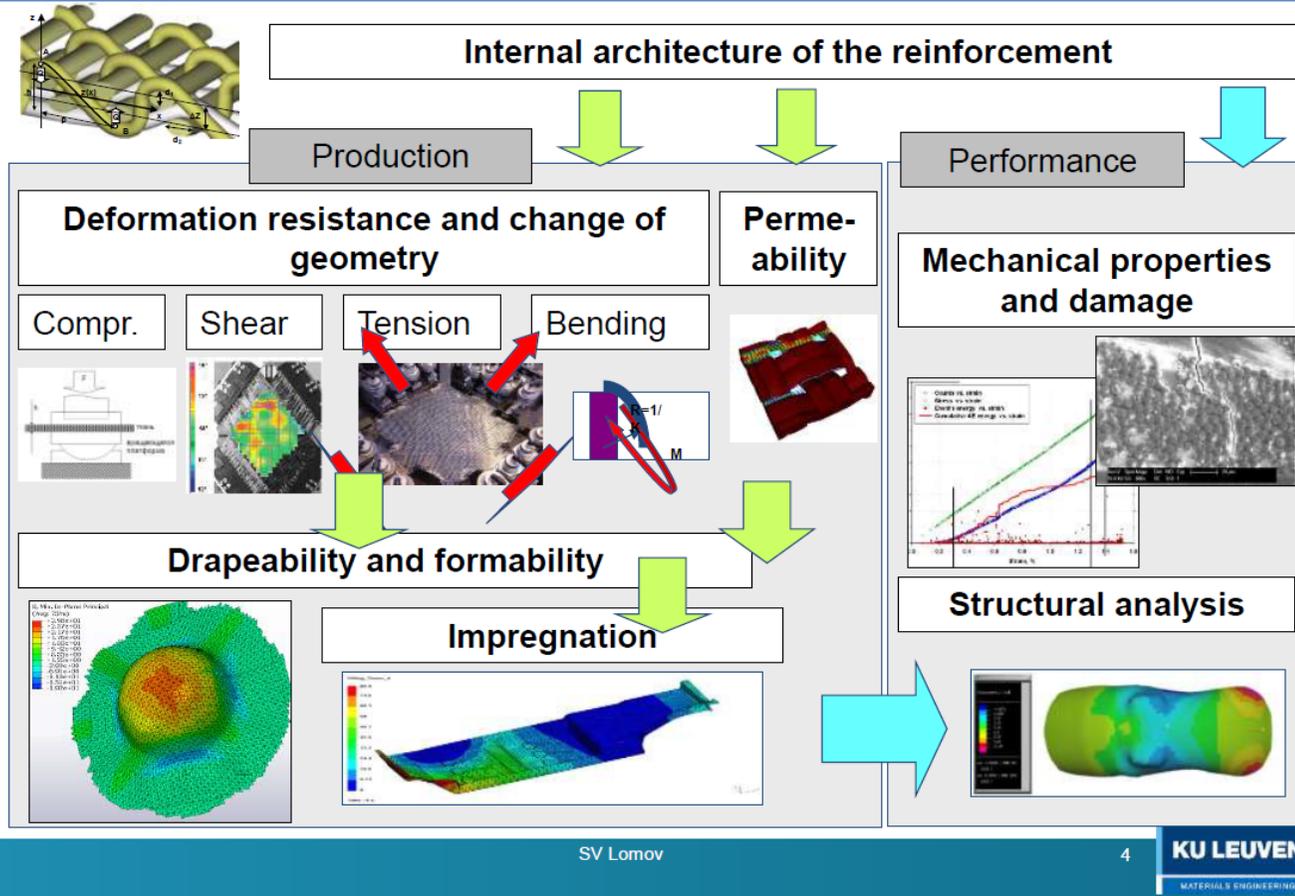
<https://filecr.com/windows/esacomp/?id=13436849765>

Les composites textiles



Les composites textiles

Integrated Design Tool: textile composites



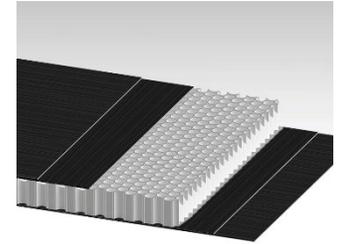
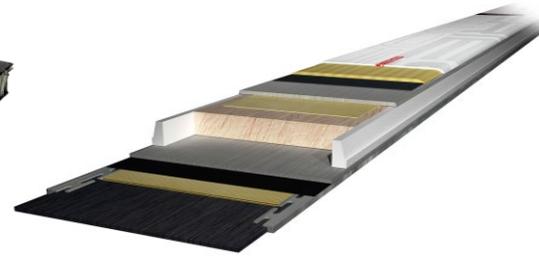
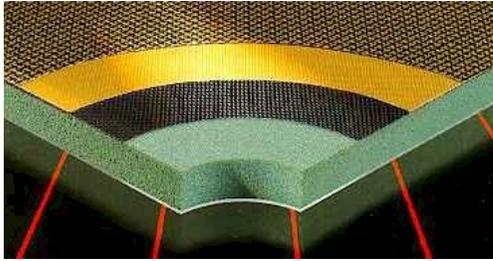
SV Lomov

4

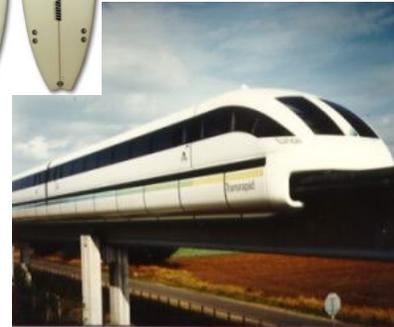
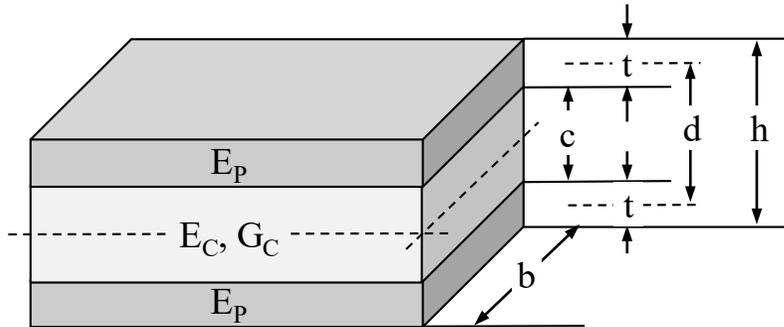
KU LEUVEN
MATERIALS ENGINEERING

<https://www.mtm.kuleuven.be/onderzoek/scalint/Composites/software/wisetex>

Structures sandwich



econcore
economic core technologies



$$D \approx D' = E_P \frac{b t d^2}{2}$$



Révision et examen

<https://docs.google.com/spreadsheets/d/1vMpbdes4FIINZgwfZNScfrZTd3DPYVLIwxQKzwLknu0/edit#gid=0>

MSE 340 Examen 30-31 janvier 1er février 2023

Fichier Édition Affichage Insertion Format Données Outils Extensions Aide Dernière modification il y a quelques secondes

75% 75% € % 0.00 123 - Calibri 11 B Z G A

| | A | B | C | D | E | F | G | H | I | J | K |
|----|---|----|----|-------|---|---|---|-----|---|--------|---|
| 1 | 32 MX avec un exa le 2 et 3 | | | | | | | | | | |
| 2 | 24 GM | | | | | | | | | | |
| 3 | Heures de passage | | | | | | | Nom | | SCIPER | |
| 4 | vous serez invité à l'heure de votre passage, térez un set de question (une sur les constituants, une sur la mise en oeuvre et une sur la mécanique des composites) aurez 5 minutes de préparation puis 15 minutes pour répondre | | | | | | | | | | |
| 5 | Lundi 30 Janv pour les MX | | | | | | | | | | |
| 6 | 1 | 8 | 15 | | | | | | | | |
| 7 | 2 | 8 | 35 | | | | | | | | |
| 8 | 3 | 8 | 55 | | | | | | | | |
| 9 | 4 | 9 | 15 | | | | | | | | |
| 10 | 5 | 9 | 35 | | | | | | | | |
| 11 | 6 | 10 | 15 | pause | | | | | | | |
| 12 | 6 | 10 | 35 | | | | | | | | |
| 13 | 7 | 10 | 55 | | | | | | | | |
| 14 | 8 | 11 | 15 | | | | | | | | |
| 15 | 9 | 11 | 35 | | | | | | | | |
| 16 | 10 | 11 | 55 | | | | | | | | |
| 17 | 11 | 13 | 15 | | | | | | | | |
| 18 | 12 | 13 | 35 | | | | | | | | |
| 19 | 13 | 13 | 55 | | | | | | | | |
| 20 | 14 | 14 | 15 | | | | | | | | |
| 21 | 15 | 14 | 35 | | | | | | | | |
| 22 | 16 | 14 | 55 | pause | | | | | | | |
| 23 | 16 | 15 | 35 | | | | | | | | |
| 24 | 17 | 15 | 55 | | | | | | | | |
| 25 | 18 | 16 | 15 | | | | | | | | |
| 26 | 19 | 16 | 35 | | | | | | | | |
| 27 | 20 | 16 | 55 | pause | | | | | | | |
| 28 | 20 | 17 | 15 | | | | | | | | |
| 29 | 21 | 17 | 35 | | | | | | | | |
| 30 | 22 | 17 | 55 | | | | | | | | |
| 31 | Mardi 31 Janv pour le solide des MX et les GM | | | | | | | | | | |
| 32 | 24 | 8 | 15 | | | | | | | | |
| 33 | 25 | 8 | 35 | | | | | | | | |
| 34 | 26 | 8 | 55 | | | | | | | | |
| 35 | 27 | 9 | 15 | | | | | | | | |
| 36 | 28 | 9 | 35 | | | | | | | | |
| 37 | 29 | 10 | 15 | pause | | | | | | | |
| 38 | 29 | 10 | 35 | | | | | | | | |
| 39 | 30 | 10 | 55 | | | | | | | | |
| 40 | 31 | 11 | 15 | | | | | | | | |
| 41 | 32 | 11 | 35 | | | | | | | | |
| 42 | 33 | 11 | 55 | | | | | | | | |
| 43 | 34 | 13 | 15 | | | | | | | | |
| 44 | 35 | 13 | 35 | | | | | | | | |
| 45 | 36 | 13 | 55 | | | | | | | | |
| 46 | 37 | 14 | 15 | | | | | | | | |
| 47 | 38 | 14 | 35 | | | | | | | | |
| 48 | 39 | 14 | 55 | pause | | | | | | | |
| 49 | 39 | 15 | 35 | | | | | | | | |
| 50 | 40 | 15 | 55 | | | | | | | | |
| 51 | 41 | 16 | 15 | | | | | | | | |
| 52 | 42 | 16 | 35 | | | | | | | | |
| 53 | 43 | 16 | 55 | pause | | | | | | | |
| 54 | 43 | 17 | 15 | | | | | | | | |
| 55 | 44 | 17 | 35 | | | | | | | | |
| 56 | 45 | 17 | 55 | | | | | | | | |
| 57 | mercredi 1er Février pour le solide des GM (les MX ont un autre exa le 2) | | | | | | | | | | |
| 58 | 46 | 8 | 15 | | | | | | | | |
| 59 | 47 | 8 | 35 | | | | | | | | |
| 60 | 48 | 8 | 55 | | | | | | | | |
| 61 | 49 | 9 | 15 | | | | | | | | |
| 62 | 50 | 9 | 35 | | | | | | | | |
| 63 | 51 | 9 | 55 | | | | | | | | |
| 64 | 52 | 10 | 15 | pause | | | | | | | |
| 65 | 53 | 10 | 35 | | | | | | | | |
| 66 | 54 | 10 | 55 | | | | | | | | |
| 67 | 55 | 11 | 15 | | | | | | | | |
| 68 | 56 | 11 | 35 | | | | | | | | |
| 69 | | | | | | | | | | | |
| 70 | | | | | | | | | | | |

| | | |
|--|--------|--------|
| Algoi Colleen | 283445 | GM |
| Aliotta Danilo | 327473 | MX |
| Ancely Emilien François | 325753 | MX |
| Arnold Melvin Matthieu | 328157 | MX |
| Aurelien Marin Nicolas Andres | 311557 | GM |
| Barrin Ramon Jose Victor | 302178 | GM |
| Bencoussan Keren Anaelle | 330878 | GM |
| Bernaert Joseph Vincent Arnaud | 302971 | GM |
| Bersier Sydney | 326773 | MX |
| Biselli Valentin | 320261 | MX |
| Bonnef Etas Maria | 311742 | MX |
| Cabannes Guilhem Benjamin Mathieu Marie | 316416 | GM |
| Cantenot Charles Hubert Emile | 299910 | GM |
| Charlot Jeanne Clothilde Suzanne | 330297 | MX |
| Cheng Laura Ying | 311022 | MX |
| Collier Clarys Adrienne | 323537 | MX |
| Correille Jehan Nivé Marc | 330381 | MX |
| Cortes Gaetan Bivente | 301753 | GM |
| De Christen Baptiste François Marie | 282216 | GM |
| Dejardin Tingyu Victor Stéphane Marie | 311966 | GM |
| Desrosiers Alain Mette | 320189 | MX |
| Desraves Olivier | 359188 | GM_ECH |
| Digonelli Léo Marco Florent | 320136 | MX |
| Evans Zoé Philomène | 323356 | MX |
| Frey Anabelle Florence | 330946 | MX |
| Garnaud Pierre Jurlilien | 353113 | GM |
| Gaudiot Benoît Thomas Charles | 275223 | GM |
| Gomes Almada Guillem Inmael | 329989 | MX |
| Gubelmann Oliviero Julien | 305501 | MX |
| Laperrière Antoine Maurice Jean | 289734 | GM |
| Lardet Armande Lucie Caroline | 313044 | MX |
| Lattion Gilles Sylvain | 330724 | MX |
| Lunven Bénédicte Marie Hélène Anne | 324254 | MX |
| Madie Gabriel Bruno Elle | 302723 | GM |
| Malmison Charles Joseph | 321745 | MX |
| Manchall Justin Marie Eric François | 311055 | GM |
| Mayence Lucy Odile Anny | 312684 | MX |
| Meyer Germain Philippe Marie | 338225 | GM |
| Nicolas Pauline Marie Constance Simone | 313020 | MX |
| PEZ Césarine | 330914 | GM |
| Polet-Villard Lucas Marius | 316179 | GM |
| Quilichini Benjamin | 287560 | GM |
| Randriamanantena Zina Arinaivo | 327092 | MX |
| Roche Luka Gabriel | 329636 | MX |
| Rosenkrantz Pia Elisa Friederike | 297074 | MX |
| Roy Arment Mathis William Clément | 289749 | GM |
| Sachot-Durette Jules Édouard Dominique Thierry | 295768 | GM |
| Servais Albin Jean Léonard | 311877 | MX |
| Soury-Lavergne Nathan François Joseph | 310858 | GM |
| Staudermann Jesse Janet | 325387 | MX |
| Studer Noah | 323146 | MX |
| Tarres Julien | 311481 | MX |
| Thiercelin Basile Corentin | 310100 | GM |
| Tran Cécile | 310872 | MX |
| Valbargiacomo Nico Dario | 325385 | MX |
| Wiederstein Cedric Christian | 312369 | MX |

| | A | B | C | D | E |
|----|---|---|---|---|---|
| 1 | Questions pour la séance de révision du 19 décembre | | | | |
| 2 | | | | | |
| 3 | Questions sur la mise en oeuvre des composites | | | | |
| 4 | | | | | |
| 5 | XXXXXX | | | | |
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| 19 | Questions sur la mécanique des composites | | | | |
| 20 | | | | | |
| 21 | yyyyy | | | | |
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